

Characterization of Films Based on Cellulose Acetate/ Poly(caprolactone diol) Intended for Active Packaging Prepared by Green Chemistry Principles

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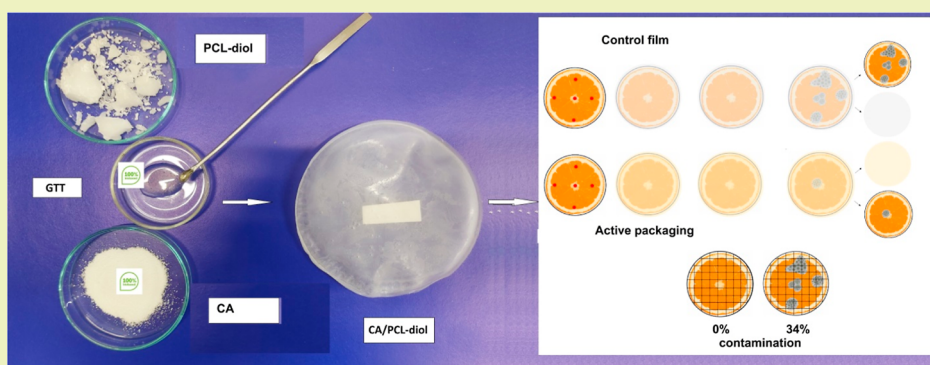
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ABSTRACT: Biodegradable active packaging films based on a cellulose acetate and poly(caprolactone diol) blend with incorporated lemongrass oil were developed. Films were prepared using a novel bio-based plasticizer, glycerol tritartarate, synthesized using the principles of green chemistry. The influence of the plasticizer, as well as the essential oil amount, on the structural, surface, mechanical, and thermal properties of the blend was investigated. The plasticizer was shown to work as a compatibilizer for two polymers, according to the results of scanning electron microscopy and surface energy analysis. Blends with a greater amount of plasticizer possessed better mechanical properties but showed worse resistance to water. The antimicrobial property of the blend with lemongrass oil was found to be superior to that of the blend without essential oil. The incorporation of lemongrass oil into the polymer blend resulted in one more step longer thermal degradation process. The optimal film properties, biodegradability, cost-effective preparation method, and additional functions made these films suitable for the production of packaging for grapefruit.

KEYWORDS: cellulose acetate, poly(caprolactone diol), glycerol tritartarate, lemongrass oil, blends, biodegradable active packaging, torsion resistance

INTRODUCTION

The packaging industry is the largest and fastest growing consumer of synthetic conventional plastic materials, which are highly dependent on fossil resources. Also, this industry, especially single-use plastics intended for food and beverages, is the main source of supplying waste plastics into the environment at an alarming rate creating a problem of microplastics, as polymers do not degrade but break down into smaller pieces, ending up in air, soil, water, as well as in living organisms.^{1–3} This has created a need for the development of green, abundant, and economic alternatives to petrochemical plastic-packaging materials such as bioplastics. The bioplastic industry is reported to be a young, innovative sector possessing great ecological and economic potential for a low-carbon, circular economy that utilizes resources more efficiently.⁴ The main functions of food

packaging have been the same throughout history: hygiene and protection during transportation and storing. Modern technology and materials have introduced new functionalities to packaging because the selection and development of packaging materials can be a partial solution to the complex problems of food safety and reduce unnecessary food wastes.⁵ Many terms are used to describe packaging with new functions, and one of them is active packaging, which involves the use of

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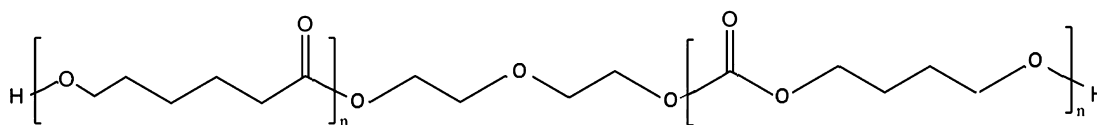


Figure 1. Chemical structure of PCL-diol.

polymers that act as supports for various active compounds that can be incorporated during the packaging manufacturing process.^{6,7} Active packaging provides extraordinary as well as vital functions to food products with an extended shelf life while ensuring their quality, safety, and integrity.^{8,9} Active green food packaging will be one answer to the global challenges related to the desperate quest for carbon neutrality, food safety, and food saving, as well as renewable materials and technologies.⁵ Sustainable active packaging materials should be tailored to impose a lower impact on the health of consumers as well as the environment. Food-packaging materials can be enriched with antimicrobials, which affect the growth of pathogens in food. These active agents slowly release from the packaging material and dissolve onto the surface of the food through direct contact, extending the shelf life and organoleptic properties of the food product.¹⁰ Huge amounts of food spoiled by microbial contamination represent one of the essential problems in food distribution and storage process. For example, fungi are responsible for the loss of about 20% of the yield of fruits and vegetables globally, which is enough food to feed millions of people. It is estimated that at least 125 million tons of fruits and vegetables are manually contaminated in Europe, with an increasing trend every year, which indicates the importance of this problem.¹¹ Natural compounds obtained from plants represent an alternative and eco-based replacement for antibiotics because these do not contribute to antimicrobial resistance, are needed in small quantities, and are obtained from different types of waste.¹² Essential oils, with recognized antimicrobial properties, can be used for this purpose as natural additives in packaging applications.¹³

Cellulose derivatives are usually most relevant for packages that come into direct contact with food. Cellulose acetate (CA) is a low-cost thermoplastic biopolymer widely used for primary food-packaging applications as a rigid wrapping film, and it has some inherent drawbacks. To counter its drawbacks, blending CA with other biopolymers can be a solution for achieving environmentally friendly, low-cost packaging materials with improved properties. In order to improve mechanical properties and processability, CA has been blended with biodegradable poly(caprolactone) (PCL), which has been used as a route to make processable food-packaging materials. There are several reports, in which CA is blended with PCL by dissolving polymers in acetone without and with a compatibilizer (polyethylene-*graft*-glycidyl methacrylate).¹⁴ The obtained results have showed an improvement in the mechanical and thermal properties of blends after the addition of a compatibilizer. El-Naggar et al. have synthesized an eco-friendly antimicrobial film based on CA and PCL loaded with copper nanoparticles for food-packaging applications. The produced films have displayed good mechanical properties and excellent air permeability when compared with films formed without copper nanoparticles.¹⁵ In a number of studies, PCL-based fibers were electrospun with CA fibers to improve the wetting properties and broaden the field of application, such as in biomedicine. A group of authors has prepared microfibers from CA blended with PCL loaded with different essential oils

for potential application in biomedical fields. The initial idea for the combination of CA with PCL was to improve CA processing properties, flexibility, and durability. A group of authors has prepared microfibers from CA blended with PCL loaded with different essential oils for potential application in biomedical fields. With the same intention, CA has been mixed with other polymers with plasticizing effects, such as poly(ethylene glycol),¹⁶ poly(ethylene succinate),¹⁷ poly(vinyl acetate),¹⁸ poly(*N*-vinyl pyrrolidone),¹⁹ polyaniline,²⁰ and so forth. Different authors have investigated the plasticizing effect of low-molecular-weight substances such as biocompatible triacetin, tripropionin, triethyl citrate, tributyl citrate, and tributyl 2-acetyl citrate²¹ as substitutes for toxic phthalates. Meier and coauthors have investigated the structural, thermal, and mechanical properties of CA films plasticized with different amounts (up to 50 wt %) of low-molecular-weight poly(caprolactone triol).²² However, researching the available literature, no study has been found about biodegradable polymer films based on CA and polyester poly(caprolactone diol) (PCL-diol) plasticized with a fully bio-based, cost-effective plasticizer prepared from glycerol and tartaric acid (TA)—glycerol tritartarate (GTT), which also proved to be a good compatibilizing agent, improving the mixing of polymers in the blend. In our view, it should be possible to obtain films based on CA, PCL-diol, and GTT using a cost-effective, solution-casting method, intended for active packaging in the food industry.

Therefore, the aim of this study was to prepare CA/PCL-diol blends with different amounts of GTT and active packaging based on the obtained material using a pure lemongrass essential oil as an antimicrobial agent. The initial idea for blending CA with PCL-diol was to achieve brittleness reduction of CA, starting from the premise of good miscibility of these polymers considering their solubility in acetone and the presence of hydroxyl groups in their structure. CA exhibits better miscibility with PCL-diol than with PCL; the formation of films with good properties based on CA and PCL is not possible without a compatibilizer, which has been confirmed by the available literature and preliminary testing. Preliminary testing was also performed using a neat CA with different amounts of synthesized plasticizer, but the mechanical and barrier properties of obtained films were not satisfying. Obtained blends prepared in a green chemistry framework were thoroughly studied in terms of their structural, mechanical, thermal, and barrier properties and applied for packaging of grapefruits, considering the susceptibility of this fruit to fungi. The novel plasticizer has also demonstrated the function of a compatibilizer. In this study, the method for the determination of the angle of torsion is applied for the first time in order to investigate the mechanical properties of films during twisting, which is a frequent deformation of packaging during manipulation.

EXPERIMENTAL SECTION

Materials. CA ($M_n \sim 50\,000$ g/mol, average acetyl content of 39.7 wt.%), poly(caprolactone diol) ($M_n \sim 2000$ g/mol) (Figure 1), and

TA were purchased from Sigma Aldrich (USA). Acetone (pro analysis) and sulfuric acid (pro analysis), glycerol, and dimethyl sulfoxide (DMSO) were supplied by ZORKA Pharma (Serbia).

Preparation of Plasticizer/Compatibilizer. The plasticizer was prepared by the esterification reaction between glycerol and TA in a molar ratio of 1:3. In order to gain a great yield of triglyceride, TA was added in three steps to glycerol. After the dissolution of TA in a minimum amount of water, one-third of the solution mass was added to the preweighed total mass of glycerol. The reaction of esterification between TA and glycerol took place for 120 min at 110 °C, while the second and third of the TA solution mass were added over 80 min. Sulfuric acid was used as a catalyst (3 drops). The chemical structure of the plasticizer is shown in Figure 2.

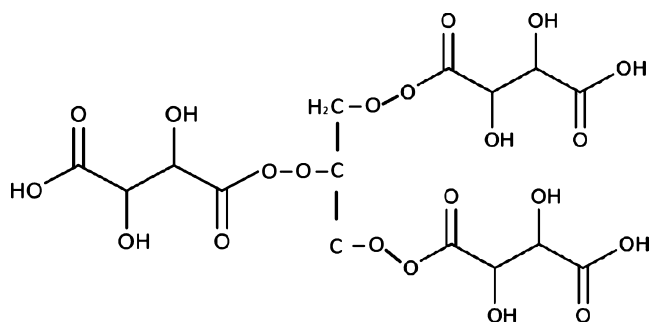


Figure 2. Chemical structure of plasticizer GTT.

Preparation of Lemongrass Oil. Lemongrass (*Cymbopogon citratus*) was grown at a greenhouse located in the village Banatska Topola (45°40'N; 20°27'E; altitude 78 m). Aboveground parts were harvested in July, dried in shade, and subjected to steam distillation. Briefly, 100 kg of dry plant material was placed in a distillation vessel ($V = 0.8 \text{ m}^3$), routed upwards through a plumbing system, and supplied with steam. After 20 min, the condensed vapor has started to collect in the Florentine flask. After 3 h, the distillation process was over. The accumulated essential oil was decanted from the water phase, dried over sodium sulfate, and stored in amber glass bottles at 4 °C until further analysis. The main compounds of lemongrass oil are nerol, citral, nonan-4-ol, camphene, 6-metil-hept-5-en-2-one, citronellal, β -caryophyllene, citronellol, caryophyllene oxide, γ -murolole, limonene, geranyl acetate, and geranial.²⁵

Preparation of CA/PCL-Diol Blends. Blends were prepared by solution blending of CA and PCL-diol in a weight ratio of 1:1. After the dissolution of CA (8 wt %) in acetone, PCL-diol was added to the solution. The mixture was homogenized at 45 °C for 2 h at a magnetic stirrer (1200 rpm), with and without the addition of plasticizer/compatibilizer, which was added in different amounts per total polymer weight (Table 1), to investigate its influence on compatibilization of polymers, as well as structural, mechanical, and applicative properties of obtained blends. One series of samples with the same amount of plasticizer/compatibilizer was prepared by variation of the lemongrass oil amount in order to get films for active packaging with an optimized composition that demonstrates antimicrobial and antioxidant properties. After homogenization of CA, PCL-diol, and plasticizer/compatibilizer in acetone, lemongrass oil was added in an amount of 3, 5, and 7 wt % (counted on the total mass of polymers in a blend). Solution-casting method was employed for the preparation of films.

FTIR Analysis. The chemical structure of blends and a plasticizer was analyzed using a Fourier-transform infrared (FTIR) spectrometer (Thermo Fisher Scientific, MA, USA) operating in total reflectance attenuation (ATR) mode. All spectra were recorded in the spectral range 4000–400 cm^{-1} , at a resolution of 4 cm^{-1} .

Scanning Electron Microscopy Analysis. The microstructure of polymer blends was investigated using a scanning electron microscope (JEOL JSM-6460, Japan) with an accelerating voltage of 20 kV. Samples were immersed in liquid nitrogen, cut, gold-sputter

Table 1. Composition of Polymer Blends

sample	amount of GTT (wt %)
CA/PCL-diol 0% ^a	0
CA/PCL-diol 5% ^b	5
CA/PCL-diol 7% ^c	7
CA/PCL-diol 10% ^d -control	10
CA/PCL-diol 10% ^e 3% LG	10
CA/PCL-diol 10% ^f , 5% LG	10
CA/PCL-diol 10% ^g , 7% LG	10
CA/PCL-diol 12% ^h	12

^aBlend with a weight CA/PCL-diol weight ratio of 1:1 without plasticizer/compatibilizer. ^bBlend with a weight CA/PCL-diol weight ratio of 1:1 with 5 wt % of plasticizer/compatibilizer (per blend weight). ^cBlend with a weight CA/PCL-diol weight ratio of 1:1 with 7 wt % of plasticizer/compatibilizer (per blend weight). ^dBlend with a weight CA/PCL-diol weight ratio of 1:1 with 10 wt % of plasticizer/compatibilizer (per blend weight). ^eBlend with a weight CA/PCL-diol weight ratio of 1:1 with 10 wt % of plasticizer/compatibilizer (per blend weight) and 3 wt % of lemongrass oil (per polymer blend weight). ^fBlend with a weight CA/PCL-diol weight ratio of 1:1 with 10 wt % of plasticizer/compatibilizer (per blend weight) and 5 wt % of lemongrass oil (per polymer blend weight). ^gBlend with a weight CA/PCL-diol weight ratio of 1:1 with 10 wt % of plasticizer/compatibilizer (per blend weight) and 7 wt % of lemongrass oil (per blend weight). ^hBlend with a weight CA/PCL-diol weight ratio of 1:1 with 12 wt % of plasticizer/compatibilizer (per blend weight).

coated, and investigated under different magnifications using the SCD-005 (Bal-tec/Leica, Wetzlar, Germany) device.

Surface Property Analysis. The surface energy of the films was investigated using contact angle experiments, performed by the sessile drop method, using two liquids (water and glycerol) at room temperature ($24 \pm 1 \text{ }^\circ\text{C}$), according to the ISO 15989 standard.²⁴ After the deposition of the tested liquid on the film surface, the contact angle was determined by the analysis of an image using the Drop Shape Analyzer KRÜSS DSA25. The results of contact angle were reported as the average value of five measurements for each sample. The surface energy of each sample can be calculated using the following equations proposed by Owens and Wendt²⁵

$$\gamma_{L1} \frac{1 + \cos \theta_1}{2} = (\gamma_{L1}^d \cdot \gamma_s^d)^{1/2} + (\gamma_{L1}^p \cdot \gamma_s^p)^{1/2} \quad (1)$$

$$\gamma_{L2} \frac{1 + \cos \theta_2}{2} = (\gamma_{L2}^d \cdot \gamma_s^d)^{1/2} + (\gamma_{L2}^p \cdot \gamma_s^p)^{1/2} \quad (2)$$

$$\gamma_s^p + \gamma_s^d = \gamma_s \quad (3)$$

where θ is the contact angle between the film and liquid drop, γ_s is the surface energy of polymer films, and γ_L is the surface energy of the testing liquid, which represents the sum of surface energy due to dispersive interactions (γ_s^d , γ_L^d) and surface energy due to polar interactions (γ_s^p , γ_L^p). Distilled water was used as a testing liquid 1, whose corresponding dispersion and polarity compound values are 21.8, 51 mN m^{-1} ; glycerol was used as a testing liquid 2, whose surface energy value due to dispersive interactions is 37 mN m^{-1} , and surface energy value which comes from polar interactions is 26.4 mN m^{-1} .

Moisture Content and Total Soluble Matter Analysis. For determination of the moisture content (MC) and total soluble matter (TSM), film samples were cut into square shapes with dimensions $20 \times 20 \text{ mm}$, weighted, and dried at $105 \pm 2 \text{ }^\circ\text{C}$ until they reached a constant weight. The values of MC for each sample were calculated using the following eq 4²⁶

$$\text{MC} = \frac{m_0 - m}{m_0} \cdot 100\% \quad (4)$$

where m_0 is the initial mass of the film sample and m is the mass of the sample after drying until reaching a constant weight. The testing was carried out in three independent measurements, using the two samples, and the results were reported as an average value. In order to determine the solubility of samples, after drying until the constant weight was reached at 105 ± 2 °C, the specimens were weighted (m'), put in 50 ml of distilled water, and kept at room temperature for 24 h, with periodic stirring. After 24 h, the samples were taken out, dried at 105 ± 2 °C until a constant weight was reached, and the measured mass was denoted as a m'' . TSM was calculated using the following eq 5

$$\text{TSM} = \frac{m' - m''}{m'} \cdot 100\% \quad (5)$$

Barrier Properties—Water Vapor Transmission Rate. The water vapor transmission rate (WVTR) for polymer films was determined according to the ISO standard,²⁷ which proposes a temperature of 25 ± 1 °C and a relative humidity of $90 \pm 2\%$. The results were averaged on three independent measurements.

Thickness. The film thickness was measured using a micrometer (Digico 1, Tesa, Swiss Made, Renens, Switzerland), with a sensitivity of 0.001 mm, at nine positions on two films from each sample, and the results were reported as the average value. The prepared films with an obtained average thickness were used for the packaging formation.

Mechanical Properties. Tensile strength and elongation at break of polymer blends were investigated using the Instron Universal Testing Instrument model 4301 (Instron Engineering Corp., Canton, MA). The samples were prepared and tested according to the ASTM standard D882-18, which implies a rectangular shape of specimens with 80 mm in nominal length and 15 mm in width. All tests were carried out at a temperature of 23 ± 2 °C, with the initial grip separation set at 50 mm and the crosshead speed set at 50 mm/min. The results were averaged on four independent measurements, and the extreme values were excluded.

To determine the prepared materials' torsion resistance, a novel torsion angle testing was devised. It was performed on a standard tensile testing machine, where the $60 \times 20 \times 0.17$ mm specimens were clamped in the fixed top grip (Figure 3). The other side of the



Figure 3. Measurement of the angle of torsion.

specimen was clamped in the bottom grip, which had a weight of 1.5 kg, which was left loose in terms of torsion. Its bottom part was left within the testing machine attachment ring. The bottom grip was manually turned until the specimen suffered a fracture. When this occurs, the bottom grip falls inside the attachment ring and the angle is subsequently measured using a digital angle measurement device. The measurements were performed at room temperature. Three specimens of the each sample set were tested, and an average value was reported.

Differential Scanning Calorimetry. The thermal properties of polymer blends were investigated using a differential scanning calorimeter (Q20 TA Instruments). Calibration was performed using indium. The samples were analyzed under a nitrogen atmosphere at a gas flow of 50 mL/min, in the temperature range

from 5 to 250 °C, at a heating rate of 10 °C min^{-1} . The glass transition temperature was determined as an inflection point.

Thermogravimetric Analysis. Thermal degradation of polymer films was investigated using the LECO 701 thermogravimetric analyzer. Measurements were carried out under the air flow (50 ml/min) in the temperature range from 25 to 800 °C at a heating rate of 10 °C/min.

X-ray Diffraction. The phase analysis of polymer blends was investigated using the Philips PW1820 X-ray diffractometer, using $\text{CuK}\alpha$ radiation (experimental setup of 30 kV and 30 mA). The data cover the 10 – 40° angular range. The degree of crystallinity was estimated using the Hermans–Weidinger method²⁸ and the MDI Jade software package. The distance between polymer chains in the amorphous phase ($\langle R \rangle$) for polymer blends was calculated using eq 6,²⁹ considering the position of amorphous halo

$$\langle R \rangle = \frac{5}{8} \left(\frac{\lambda}{\sin \theta} \right) \quad (6)$$

where λ (0.154 nm) is the wavelength of the radiation, and 2θ is an angle of diffraction.

Investigation of Antimicrobial Activity. For test, microorganisms were chosen as referent strains of food-borne bacteria: *Escherichia coli*, *Salmonella* Typhimurium, *Listeria monocytogenes*, *Staphylococcus aureus*, as well as yeasts and fungi references: *Saccharomyces cerevisiae*, *Candida albicans*, and *Aspergillus brasiliensis*. All strains were obtained from American Type Culture Collection (ATCC) and stored at -72 °C. Prior to use, microorganisms were incubated at Plate Count Agar (PCA, HiMedia, Mumbai, India) or Sabouraud Dextrose Agar (SDA, HiMedia, Mumbai, India), respectively.

In order to assess of antimicrobial activity of pure essential oil, the disk-diffusion method reported in detail by Riabov et al.³⁰ was selected. Briefly, freshly prepared suspensions of each microorganism (approx. $6 \log$ CFU/mL) were used, while the tested volume of LG of $10 \mu\text{L}$ was applied into sterile cellulose-based disks in triplicate. After the incubation period, the inhibition effect was measured as diameters of the obtained zone around disks. The obtained results based on diameters of the inhibition zone can be interpreted as follows: <22 mm = resistant; 22 – 26 mm intermediate; and >26 mm sensitive microorganism on the tested sample. The sterile distilled water was a negative control, while antibiotic (chloramphenicol (30 mcg)) and antimycotic (3% cycloheximide) were positive controls. Additionally, minimal inhibitory concentration (MIC) for all sensitive microorganisms was determined by microdilution methods³¹ using original LG oil (as 100%), and a series of dilutions were prepared in 5% DMSO (50, 25, 12.5 ... 0.195, 0.0975% v/v). In order to evaluate the pharmacodynamic potential of the antimicrobial effect of the determined MICs, the time-kill study was done. Aćimović et al.³² have described in detail time-based monitoring of the antimicrobial effect of MICs during contact with sensitive microorganisms. Contact times were maximum of 24 h for bacteria and 72 h for yeast and fungi, while sampling was done in 10 points during the mentioned period. Comparatively, non-treated bacterial suspension was incubated under the same conditions.

In order to determine the antimicrobial effect during direct contact of microbial cells and the prepared CA/PCL-diol 10% films (control and 3, 5, 7% LG addition), the previously mentioned disk diffusion was selected, but with one main modification. Instead of sterile cellulose-based disks (for application of essential oil), prepared films are cut in the shape of a disk of the same diameter (6 mm) as the commercial disks and applied on the inoculated nutrient media. All analyses were done in triplicate.

Preliminary Assessment of Fresh Fruit Packaging Efficiency.

In order to assess the efficiency of biological activities of the formed blends, the grapefruit (*Citrus paradisi* Macf.) slices are packed in the improvised packaging system in such a way that the entire surface of the grapefruit slice was in direct contact with the packaging material (Figure 4). The fresh untreated and washed grape fruits are selected for the test and cut and drained under sterile conditions in order to

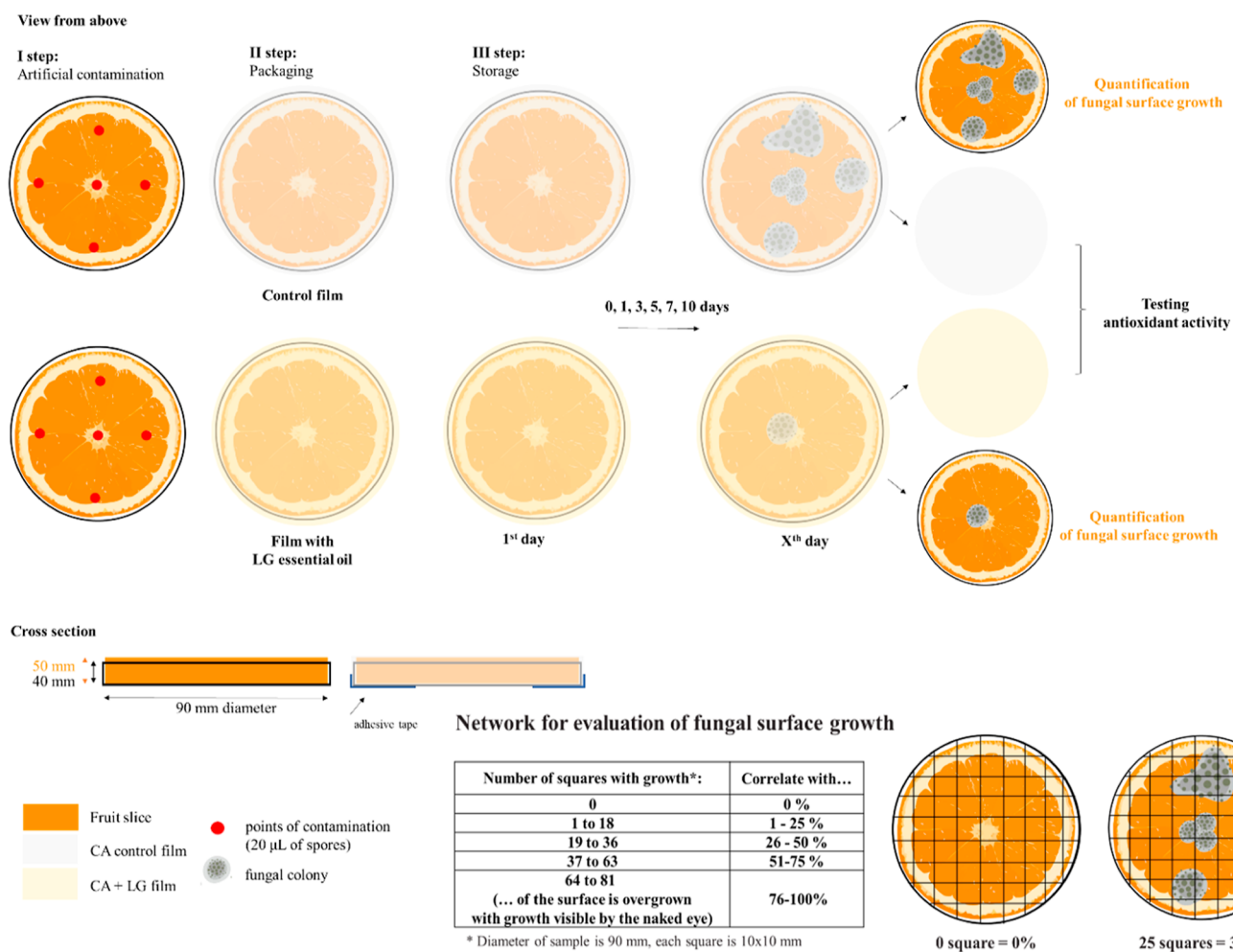


Figure 4. Experimental setup for evaluation of packaging efficiency.

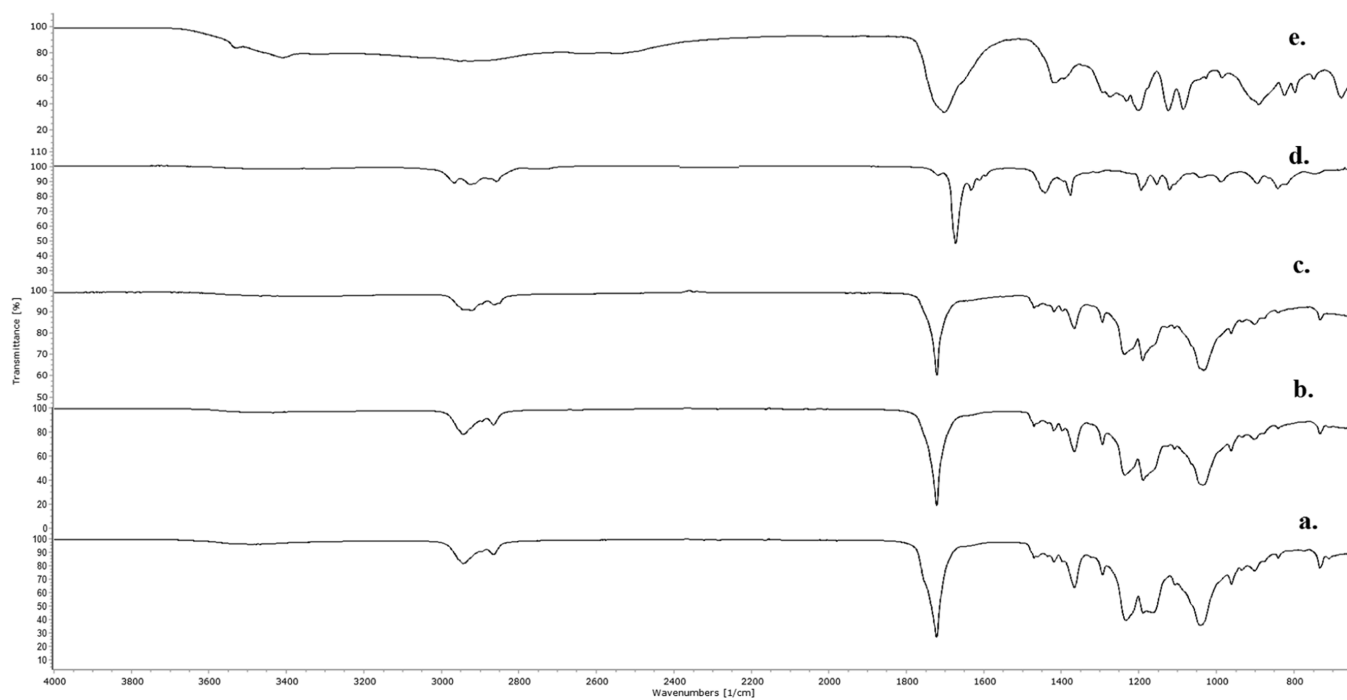


Figure 5. FTIR spectra of (a) CA/PCL 0% blend, (b) CA/PCL 10% blend, (c) CA/PCL 10%, 5% LG blend, (d) lemongrass oil, and (e) GTT plasticizer.

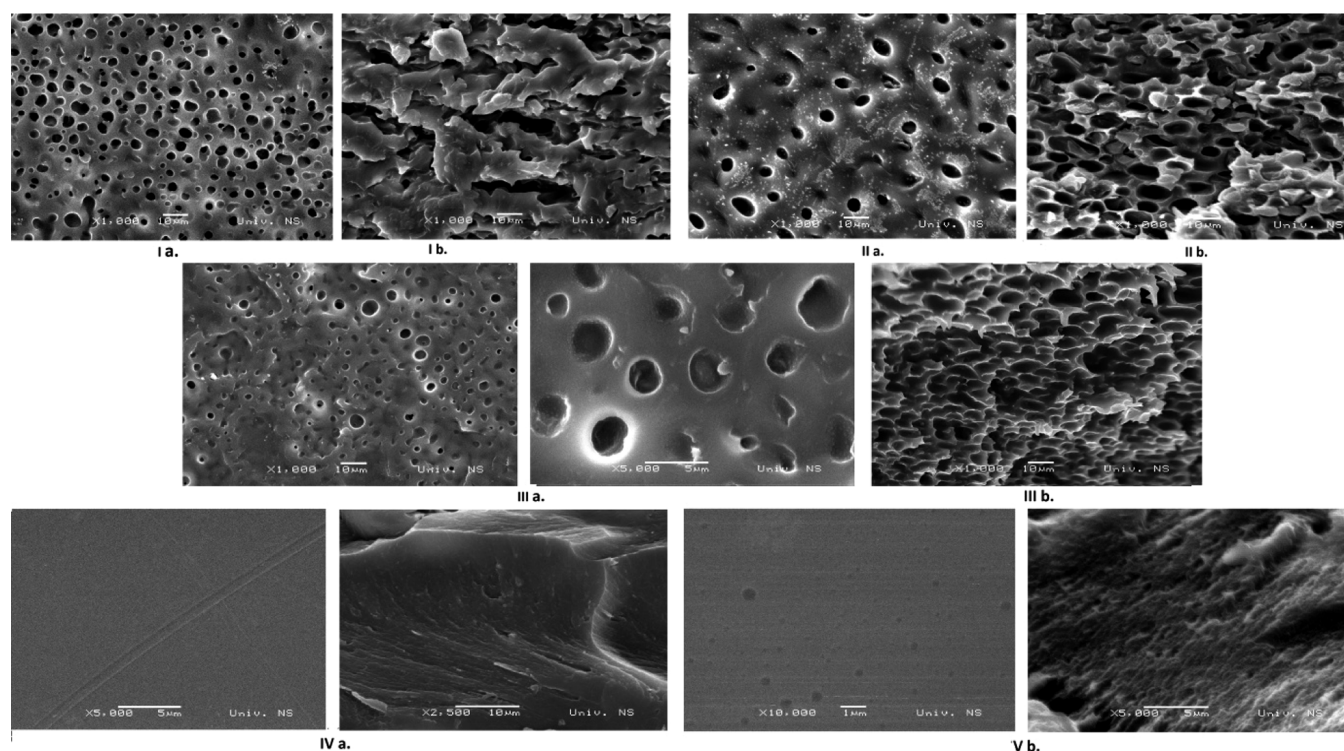


Figure 6. SEM micrographs of the top surface (a) and cross section (b): (I) CA/PCL-diol 0%, (II) CA/PCL-diol 5%, (III) CA/PCL-diol 12%, (IV) neat CA, and (V) CA 10%.

obtain equal slices (diameter 90 mm, thickness 6 mm). Each slice was put in a suitable holder (height 5 mm). After artificial contamination by *A. brasiliensis* spores (10^4 CFU/mL) using 5 points on the slice surface, CA/PCL blends (control and LG) were used for packaging of each holder with slice. The blends were placed on the contaminated surface of the grapefruit and then the maximum tightening was performed (the difference in the height between the slice and the holder allowed complete contact between the packaging and the fruit surface). The adhesive tape was used to ensure the obtained system conditions. All samples were stored at 15 ± 1 °C for 10 days, while sampling was done after 1st, 3rd, 5th, 7th, and 10th storage days. Prior to imaging, slices were unwrapped for better monitoring of fungal growth on samples. Additionally, assessment of antimicrobial treatment was done by counting the squares with fungal growth using the formed network (Figure 4). The network allows a systematic approach when the overgrown surface has to be rated by the naked eye. After unwrapping, the formed network was placed onto the grapefruit slice and counted the squares with growth in them. The number of squares with fungal growth was correlated with the percent of the visible growth after each sampling time.

RESULTS AND DISCUSSION

Results of FTIR Analysis. The FTIR spectra of polymer blends without (a and b) and with essential oil (c) as well as the spectra of pure lemongrass oil (d) and compatibilizer (e) are presented in Figure 5. A broad peak from 3600 to 3350 cm^{-1} present in the spectra of blends is assigned to the OH stretching of the CA and PCL-diol. In the FTIR spectrum of the compatibilizer, two peaks at 3410 and 3324 cm^{-1} correspond to the stretching of the secondary hydroxyl group of TA and the OH group of carboxylic acid, respectively. In the FTIR spectrum of a neat lemongrass oil, a peak attributed to the presence of an OH group appears at 3473 cm^{-1} (citronellol, nonan-4-ol). Peaks between 2900 and 2500 cm^{-1} in the spectra of blends and compatibilizer correspond to the C–H stretching in PCL-diol and compatibilizer, respectively. In the

FTIR spectra of the neat oil, peaks which correspond to C–H stretching appear at 2969, 2926, 2912, and 2857 cm^{-1} .

The peak at 1720 cm^{-1} is visible in all spectra, and it is attributed to the ester bond in CA, polycaprolactone diol chain, plasticizer, and lemongrass oil. This peak in the plasticizer spectrum confirms that the esterification reaction has been successfully performed between TA and glycerol. A strong peak at 1673 cm^{-1} is attributed to the unsaturated conjugated C=O group present in the citral. Small peaks at 1634, 1617, and 1442 cm^{-1} in the spectrum of essential oil are attributed to the C=C stretching. The band at 1470 cm^{-1} in the spectrum of blends is assigned to –CH deformation, while the band at 1420 cm^{-1} visible in the spectra of blends and compatibilizer is attributed to C–O–H bending. A strong peak at 1366 cm^{-1} in the FTIR spectra of polymer blends corresponds to C–O–H stretching in pyranose. Peaks at 1294, 1233, 1187, and 1034 cm^{-1} correspond to O–C stretching in ester groups. In the spectrum of compatibilizer, there is a peak at 1202 cm^{-1} , which is assigned to O–C stretching from the carboxylic group; two peaks at 1184 and 1085 cm^{-1} correspond to O–C and C–C–O stretching of ester groups. In the FTIR spectrum of lemongrass oil, in this range, there are peaks which correspond to the O–H stretching (1377, 1194 cm^{-1}), C=C vibrations (1155 and 1120 cm^{-1}), and stretching of the ester group (1043 cm^{-1}). A weak peak in the range from 985 to 961 cm^{-1} visible in all spectra corresponds to the C–O stretching of hydroxyl group. An expressed peak in the plasticizer spectrum at 890 cm^{-1} and the weak peaks in the range between 790 and 732 cm^{-1} of all samples correspond to C–H bending. The addition of plasticizer does not affect the changes in the FTIR spectrum (Figure 5a–c). However, the inclusion of lemongrass oil in the blend composition results in additional peaks between 2970 and 2800 cm^{-1} in the FTIR spectrum—five peaks at 2944,

2921, 2895, 2864, and 2850 cm^{-1} instead of three at 2944, 2922, and 2864 cm^{-1} in the FTIR spectrum of blends without added essential oil (Figure 5a,b).

Results of SEM Analysis. Figure 6 shows the scanning electron micrographs of the top surface (a) and cross section (b) of CA/PCL-diol blends with different amounts of plasticizer (Figure 6I–III), as well as of pure CA (Figure 6IV) and CA with the plasticizer (Figure 6V). Pure CA, with and without the plasticizer, possesses a smooth surface. The used plasticizer acts as a compatibilizer, which is evident according to the nature and number of pores. Therefore, the blend without GTA (Figure 6Ia,b) possesses the greatest number of pores, which are deep and non-uniformly distributed; the cross section demonstrates segregation of CA and PCL-diol; the formation of pores could be the consequence of the existence of voids between polymer chains of CA and PCL-diol, as a result of significantly stronger cohesion forces than adhesion ones. Both polymers are highly soluble in acetone, therefore, their chains take the conformation of a random coil that occupies space with a larger radius. Random coils of CA and PCL-diol permeate each other, and space between them after acetone evaporation creates a pore. The addition of plasticizer/compatibilizer leads to a better mixing of polymers, which results in a decrease in the pore number and depth, acting as a binding agent between CA and PCL-diol, which is particularly visible in the cross section (b), which with the addition of the plasticizer takes over a more regular honeycomb-like structure.

Surface Energy Measurements. The results of the surface energy of polymer blends give the information about miscibility of their compounds.³³ The results of contact angle measurements and calculated surface energy for the virgin polymers and polymer blends with different amounts of plasticizers are presented in Table 2. Increasing the plasticizer

Table 2. Contact Angle and Surface Energy Values for CA/PCL-Diol Polymer Blends with a Different Amount of Plasticizer/Compatibilizer

sample	contact angle ($^{\circ}$)		surface energy (mN m^{-1})		
	θ_1	θ_2	γ_s	γ_s^p	γ_s^d
CA/PCL 0%	69.4	65.1	33.67	21.62	12.05
CA/PCL 5%	67.2	64.8	35.04	24.20	10.84
CA/PCL 7%	65.3	62.1	41.41	26.53	14.89
CA/PCL 10%	59.7	56.9	42.00	28.31	12.89
CA/PCL 12%	55.8	53.2	44.41	31.04	13.37
CA/PCL 10%, 3% LG	60.2	56.3	41.12	28.94	12.18
CA/PCL 10%, 5% LG	62.4	56.1	39.80	23.72	16.08
CA/PCL 10%, 7% LG	66.2	53.3	40.06	15.36	24.70
CA 10% ^a	50.6	48.4	48.62	34.83	13.79
CA neat	56.2	53.3	44.05	30.25	13.80
PCL diol	57.03	54.6	45.83	31.92	11.91

^aCA with 10 wt % of plasticizer.

amount leads to an increase of surface energy of polymer blends, as a result of better compatibilization and hydrophilic nature of the plasticizer. Virgin polymers have similar values of surface energy, which are above the surface energy values of the polymer blends. Adding plasticizer/compatibilizer results in better wetting between polymer blends and used solvents (lowering the contact angle), which is the consequence of compatibilizer's hydrophilic nature. Therefore, the greatest value of surface energy has been recorded for the polymer

blend with 12 wt % of compatibilizer (31.04 mN m^{-1}), whose greater amount comes from the polar interactions. The addition of lemongrass oil in the blend composition does not affect surface energy significantly. However, the addition of oil in an amount of 7 wt % results in the increase of surface energy, which comes from non-polar interactions, which is expected considering the hydrophobic nature of the essential oil.

MC, Total Soluble Matter, and Water Vapor Transmission Rate Analysis. MC decreases with addition of 5 wt % of plasticizer, because of decreasing of the number of pores (results of SEM analysis) and better adhesion between two polymers in the presence of plasticizer/compatibilizer (Table 3). Further addition of the plasticizer leads to an increase in

Table 3. MC, TSM, and WVTR Values for Polymer Blends with Different Amounts of Plasticizer and Lemongrass Oil

Sample	MC (%)	TSM (%)	WVTR ($\text{g/m}^2\text{h}$)
CA/PCL-diol 0%	6.16	-6.89	106.22
CA/PCL-diol 5%	3.88	-2.79	87.08
CA/PCL-diol 7%	5.59	-2.76	66.96
CA/PCL-diol 10%	9.41	1.24	70.53
CA/PCL-diol 12%	10.33	2.98	73.45
CA/PCL-diol 10%, 3% LG	9.35	1.12	70.23
CA/PCL-diol 10%, 5% LG	9.27	1.09	71.26
CA/PCL-diol 10%, 7% LG	8.21	1.00	75.67

the MC value due to its hydrophilic nature. In a series prepared by variation of essential oil, addition of lemongrass oil does not affect the MC value at lower concentrations, but addition at a concentration of 7 wt % leads to a decrease in MC from 9.41 (for the control sample CA/PCL/diol 10%) to 8.21% (for CA/PCL 10%, 7% LG) due to the hydrophobic nature of essential oil.

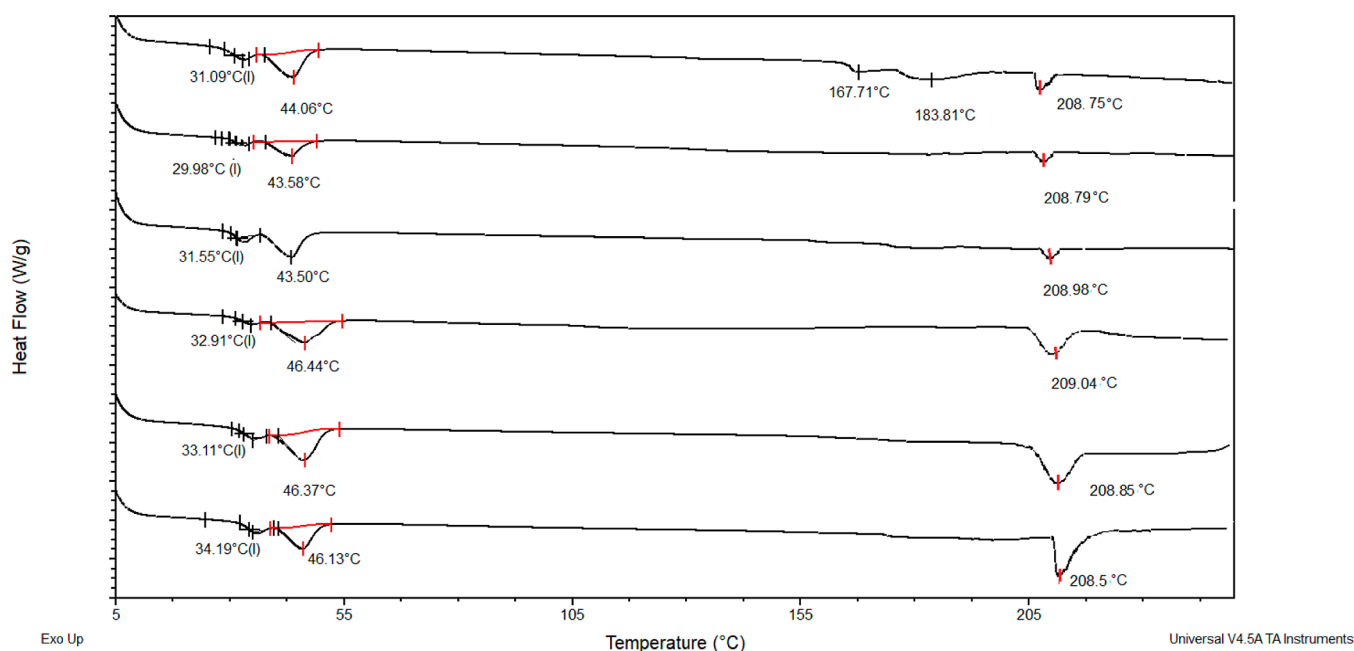
TSM values of films are especially important in packaging of food with a great amount of water such as fruits. Obtained blends with lower amounts of plasticizer (0–7 wt %) do not possess solubility in water (Table 3). Addition of the plasticizer in an amount of 10 and 12 wt % results in low solubility of polymer blends, as a consequence of the hydrophilic and water-soluble plasticizer. Lemongrass oil added in investigated amounts does not significantly affect the TSM value in comparison to the control film.

WVTR is an important parameter for food packaging, because of the significant role of water in microbial stability and reactions, which leads to the deterioration of product quality.³⁴ The blend without plasticizer possesses the greatest WVP value due to the phase segregation and porous structure. Addition of the plasticizer in an amount up to 7 wt % leads to a decrease in WVP (Table 3). Further addition of plasticizer results in a slight increase in WVP. Addition of lemongrass oil in an amount of 7 wt % results in increased WVP in comparison to the control sample. The gas permeability of the polymer is influenced by the crystalline–amorphous phase ratio, wherein the crystalline phase is impermeable to gases, and the greater amount of amorphous phase results in better gas permeability.³⁵ Lemongrass oil in an amount of 7 wt % acts as a plasticizer, increasing the mobility of polymer chains and polymer–oil interaction, which results in increasing of the WVP value.³⁶

Mechanical Analysis and Film Thickness. The thickness values for films used for packaging formation are given in the

Table 4. Results of Thickness, Tensile Strength (TS), Elongation at Break (EB), and the Angle of Torsion

sample	thickness (μm)	TS (MPa)	EB (%)	thickness (μm)	the angle of torsion (deg)
CA/PCL-diol 0%	169.3 \pm 0.013	20.31 \pm 2.56	4.82 \pm 1.28	170.0 \pm 0.012	65.9 \pm 2.40
CA/PCL-diol 5%	182.7 \pm 0.074	20.36 \pm 3.60	6.56 \pm 2.27	170.0 \pm 0.012	72.30 \pm 3.11
CA-PCL-diol 7%	156.0 \pm 0.021	23.00 \pm 4.25	9.18 \pm 2.34	170.0 \pm 0.012	79.50 \pm 2.13
CA/PCL-diol 10%	174.5 \pm 0.008	23.56 \pm 1.91	10.17 \pm 4.02	170.0 \pm 0.012	84.70 \pm 1.45
CA/PCL-diol 12%	184.1 \pm 0.007	24.84 \pm 2.68	11.72 \pm 2.52	170.0 \pm 0.012	107.40 \pm 3.16
CA/PCL-diol 10%, 3% LG	174.5 \pm 0.007	21.29 \pm 1.17	10.23 \pm 3.35	170.0 \pm 0.012	85.26 \pm 2.67
CA/PCL-diol 10%, 5% LG	175.0 \pm 0.011	21.79 \pm 3.32	10.79 \pm 3.12	170.0 \pm 0.012	87.10 \pm 1.85
CA/PCL-diol 10%, 7% LG	174.5 \pm 0.014	24.91 \pm 4.49	11.28 \pm 2.92	170.0 \pm 0.012	98.03 \pm 2.85

**Figure 7.** DSC curves of polymer blends: (a) CA/PCL-diol 0%, (b) CA/PCL-diol 5%, (c) CA/PCL-diol 7%, (d) CA/PCL-diol 10%, (e) CA/PCL-diol 12%, and (f) CA/PCL diol 12%, 7% LG.

first column of Table 4, and in the fourth column for the films subjected to the determination of torsion resistance. Tensile strength and elongation at break values are similar and increase with an increase in the compatibilizer/plasticizer amount. The addition of lemongrass oil in amounts of 3 and 5 wt % per weight of blend does not have an effect on the tensile strength, but the addition of the lemongrass oil in the amount of 7 wt % results in improved tensile strength in comparison with the control blend. The used oil is compatible with the blend, and in this amount has a function of plasticizer, which can be concluded according to the improved value of elongation at break. The angle of torsion increases with increasing plasticizer amount, which is in coherence with tensile properties. Therefore, the sample CA/PCL-diol 12% has the greatest value of angle of torsion (107.40°), while the lowest one was recorded for the blend without plasticizer (65.90°). Variation of the essential oil amount in a series with the same composition, it can be observed that the angle of torsion increases with the increasing LG oil amount, but this increase is not significant for lower amounts of etheric oil.

DSC Analysis. DSC thermograms of polymer blends are presented in Figure 7. Increasing the amount of plasticizer in polymer blends leads to the decrease of glass transition temperature—from 34 °C for the polymer blend without plasticizer to 31 °C for the polymer blend with the greatest amount of plasticizer (CA–PCL-diol 12%). The glass

transition temperature of blends is between the glass transition temperatures of virgin polymers, which are -56°C for PCL diol (Figure S1, Supporting Information) and 206°C for CA (Figure S2, Supporting Information). The first melting peak appears at 43–46 °C and corresponds to the melting of PCL-diol in the blend. The other melting peak at 208–209 °C is assigned to the melting of CA in the polymer blend. The addition of GTT does not affect significantly the glass transition temperature, leading to its decrease only by 5 °C at the greatest investigated amount (12 wt %). This has confirmed that GTT does not have a strong plasticizing effect, but acts more as a compatibilizer between polymers in blend, which is confirmed by the results of SEM and WVTR measurements. The presence of lemongrass oil in the polymer blend in an amount of 7 wt % does not affect significantly the phase transitions of the polymer blend (Figure 7f). Peaks which appear at 167 and 183 °C are attributed to the essential oil evaporation.

Thermal Gravimetric Analysis. TGA curves of the virgin polymers (CA and PCL-diol) and polymer blends with and without essential oil are presented in Figure 8, and the results obtained from the derivative thermogravimetric curve (DTG) are summarized in Table 5. The DTG curves are given in the Supporting Information (Figures S3–S6). Degradation of a neat CA and blends is carried out in three steps, where the second step is evidenced as the main thermal decomposition

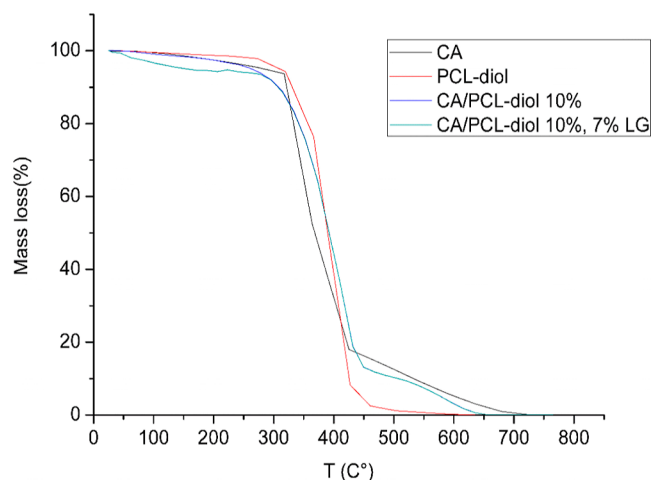


Figure 8. TGA thermograms of neat CA, PCL-diol, CA/PCL-diol 10%, and CA/PCL-diol 7% LG.

event. The first step, up to 317 °C, in the thermogram of neat CA corresponds to the mass loss of 6% due to the elimination of moisture and deacetylation.^{37,38} The second step, up to 425 °C with a mass loss of 82%, is attributed to the degradation of the CA main chain, the breakdown of the glycosidic linkages, and the formation of furanic compounds.³⁹ The third stage, up to 767 °C, corresponds to the complete degradation and sample carbonation until 0.1% of the residual mass. The TGA curve of PCL-diol displays one degradation step,⁴⁰ with a maximal degradation rate at 427 °C. The first degradation stage in the thermogram of the blend, which ends up at 277 °C, is assigned to the mass loss of 8% due to the elimination of moisture and deacetylation. In the second, main degradation stage, up to 449 °C, blend loses 87% of weight, due to the degradation of the CA main chain and breaking of the ester linkages in the PCL-diol chain. The third stage is a result of the final degradation and sample carbonation ending up with a residual mass of 0.12%. A combination of CA and PCL-diol in this ratio in the polymer blend results in a material with a little bit of thermal stability in comparison to the virgin polymers, which is manifested in the longer second and third phases. Incorporation of lemongrass oil into the polymer blend results in a four-step degradation process, where the first step up to 172 °C corresponds to degradation of lemongrass oil, moisture loss, resulting in weight loss up to 6%.⁴¹ The second step, up to 277 °C, is attributed to deacetylation. Further decomposition is carried out in a similar way to decomposition of the polymer blend without oil.

XRD Analysis. Wide-angle X-ray diffraction (WAXS) was used to investigate the level of polymer blends' microstructure arrangement. Figure 9 illustrates the XRD patterns for polymer blends with different amounts of plasticizer. The results of the crystallinity degree for polymer blends and neat polymers are presented in Table 6. In the XRD pattern of the neat PCL-diol (Figure 10), there are three intense reflections at $2\theta = 21.30^\circ$, 22.00° , and 23.60° , and among the investigated samples the

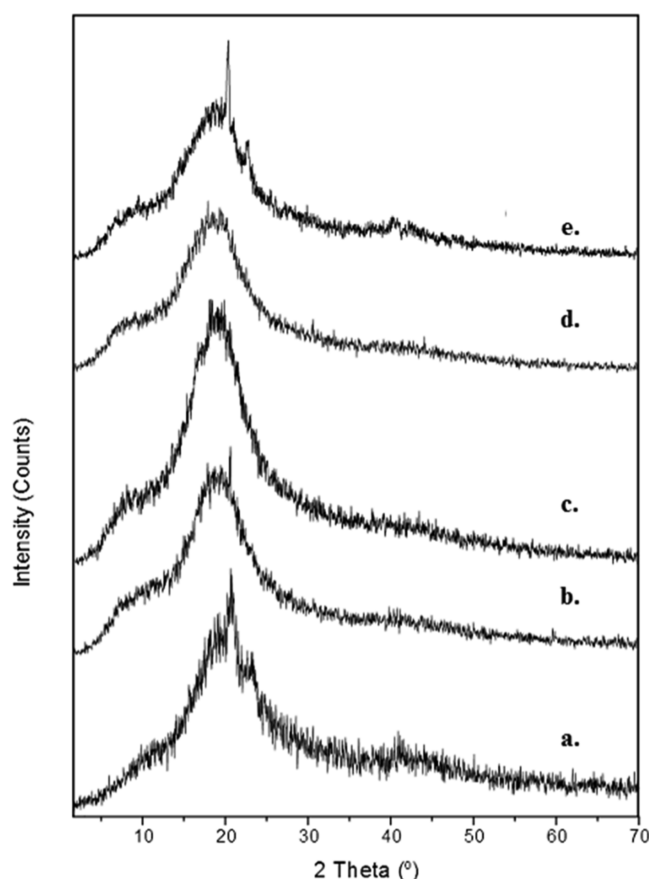


Figure 9. XRD patterns of polymer blends with different amounts of plasticizer: (a) CA/PCL-diol 0%, (b) CA/PCL-diol 5%, (c) CA/PCL-diol 7%, and (d) CA/PCL-diol 10%, CA/PCL-diol 12%.

Table 6. Degree of Crystallinity and Interchain Spacing ($\langle R \rangle$) for Polymer Samples

sample	crystallinity (%)	$\langle R \rangle$ (Å)
CA-PCL-diol 0%	21.05	5.54
CA-PCL-diol 5%	20.86	5.68
CA-PCL-diol 7%	23.20	5.60
CA-PCL-diol 10%	22.78	5.33
CA-PCL-diol 12%	22.16	5.21
CA 10%	19.70	5.46
CA neat	21.81	6.15
PCL diol	28.80	

highest degree of crystallinity is observed for the PCL-diol (28.80%). A broader reflection that appears in the diffractograms of polymer blends in the range from $2\theta = 18.80^\circ$ to $2\theta = 21.60^\circ$ is the consequence of amorphous contribution of blending with CA.²⁹ CA is a semicrystalline polymer, whose spectrum has a broad crystal peak heavily overlapped with that of primary amorphous halo. Polymer blends have a similar degree of crystallinity, in the range between 21.05 and 23.20%,

Table 5. DTG Peak Maxima Values for Neat CA, CA/PCL-Diol 10%, and CA/PCL-Diol, 7% LG

sample	TS% (°C)	T_{dmaxI} (°C)	T_{dmaxII} (°C)	$T_{dmaxIII}$ (°C)	T_{dmaxIV} (°C)	residual mass
CA	270.3	131.6	364.0	545.9		0.1
CA/PCL-diol 10%	258.3	105.7	431.6	597.1		0.12
CA/PCL-diol 10%, 7% LG	153.1	105.7	222.2	410.8	597.1	0.12

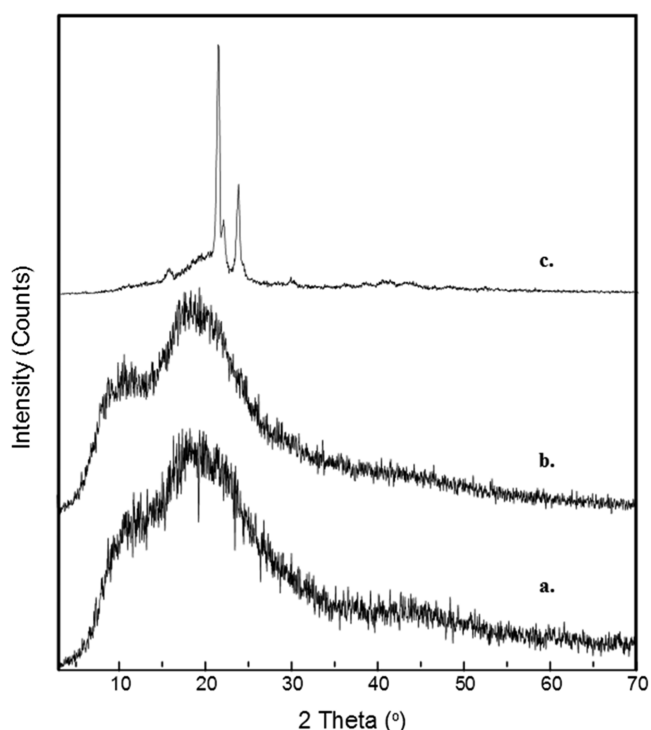


Figure 10. XRD patterns of: (a) neat CA, (b) CA 10%, and (c) neat PCL-diol.

which is in coherence with the results of DSC analysis. The average interchain spacing, $\langle R \rangle$, increases with the addition of plasticizer in the amount of 5 wt %. A further increase in the plasticizer amount leads to a decrease in the $\langle R \rangle$ value, which implies more efficient chain packing and a compatibilizing effect of the plasticizer.

Antimicrobial Properties. According to the obtained results of the disk-diffusion test (Figure 11a), LG essential oil has shown a low antimicrobial activity against both tested Gram-negative bacteria *E. coli* ATCC 25922 and *Salmonella* Typhimurium ATCC 13311. On the other hand, a very high antimicrobial activity was observed against Gram-positive bacteria *S. aureus* ATCC 25923 and *L. monocytogenes* ATCC 35152. The gained inhibition zone for these two bacteria is higher than for chloramphenicol, which indicates the strong antimicrobial potential of LG essential oil for preventing of propagation of bacterial growth in the presence of an adequate amount of oil. Furthermore, inhibition of growth in the presence of LG oil was also observed for all tested yeasts and fungi (*S. cerevisiae* ATCC 9763, *C. albicans* ATCC 10231, and *A. brasiliensis* ATCC 16404). Compared with cycloheximide as an antimycotic agent, the obtained zones are at the same or higher level in the case of LG essential oil. It can be concluded that using LG essential oil against yeast and fungal contamination is promising, but further investigation can be directed on testing a larger number of food contamination isolates, not only referent strains. For the determination of the

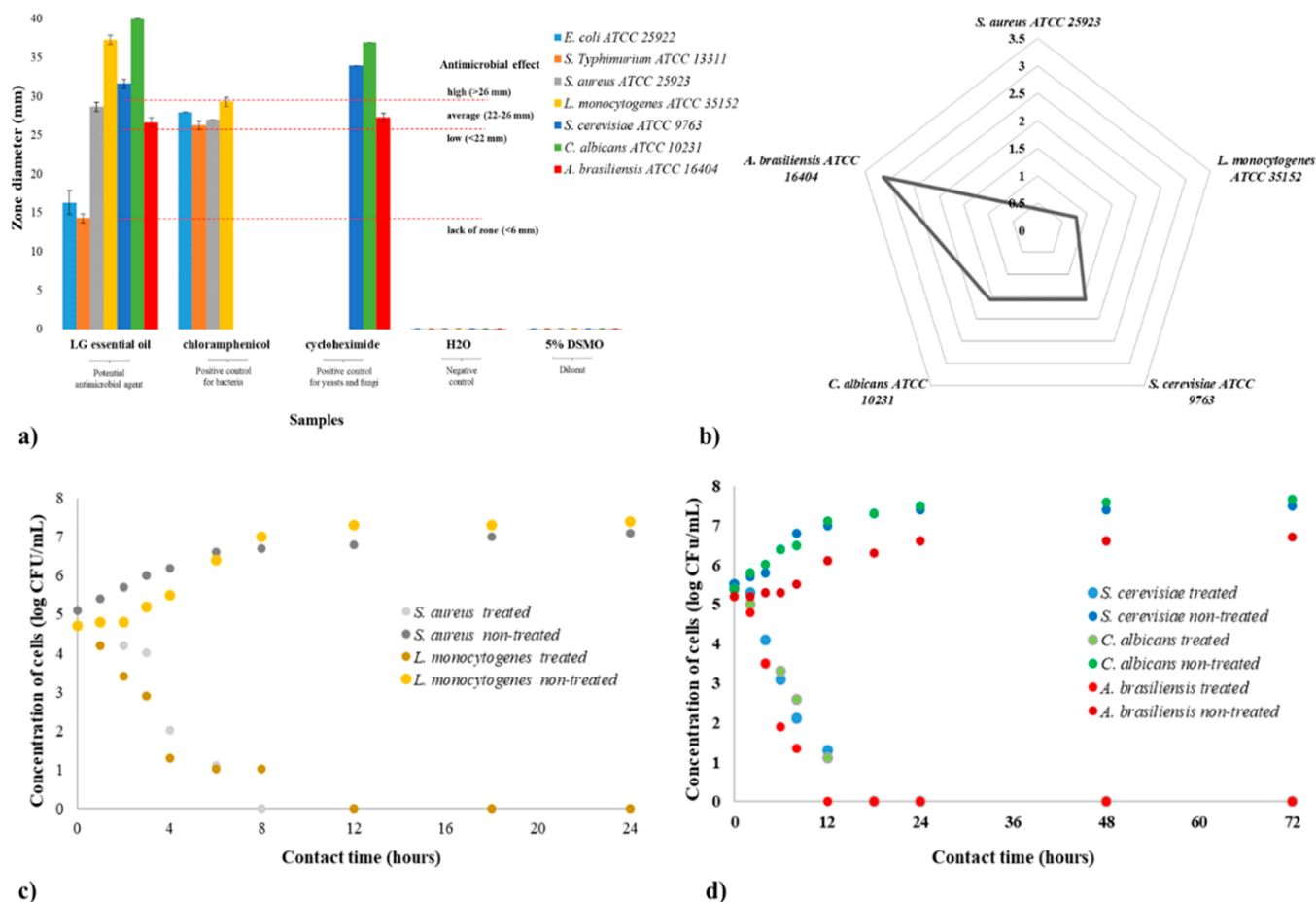


Figure 11. (a) Inhibition zone; (b) MIC (%v/v); (c) time-kill kinetics study for sensitive bacteria; and (d) time-kill kinetics study for sensitive yeasts and fungi.

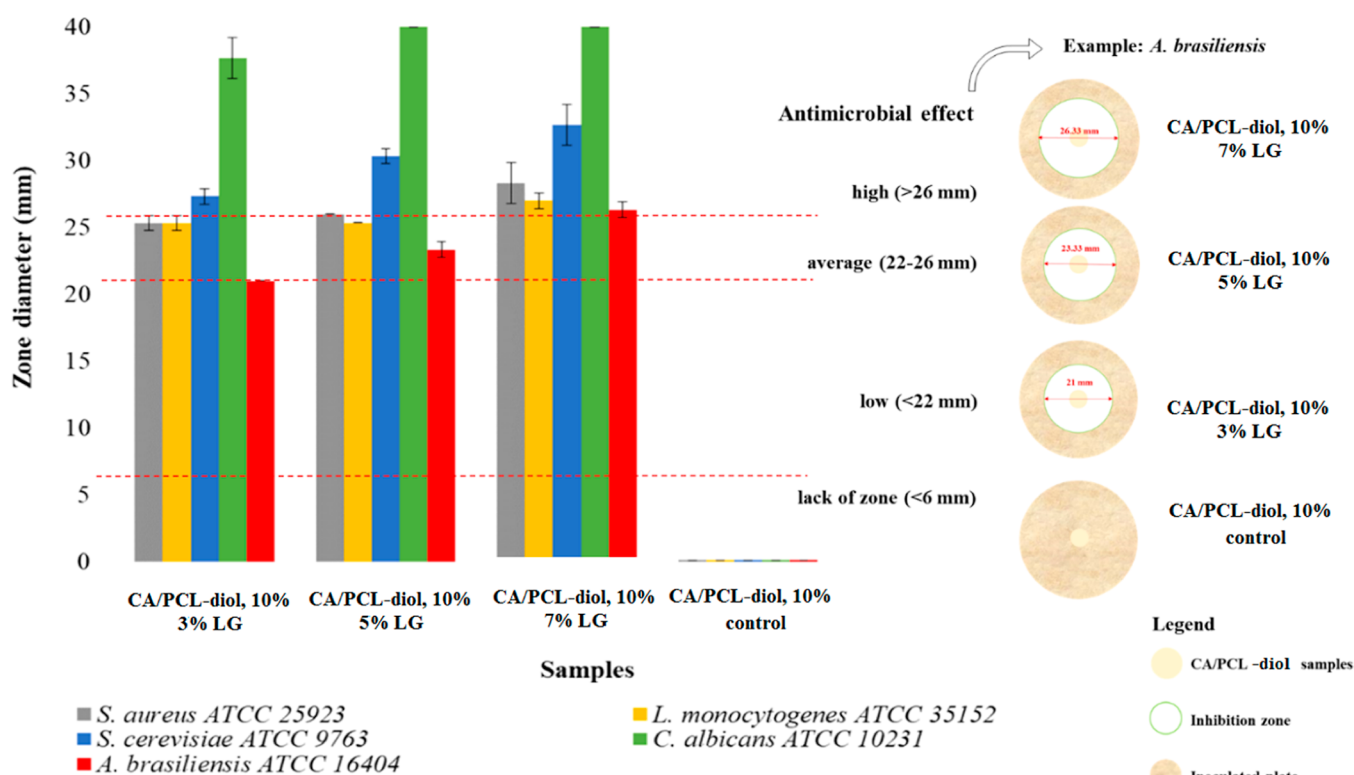


Figure 12. Antimicrobial effect of CA/PCL-diol samples.

minimal concentration of LG essential oil that inhibits the growth of the sensitive microorganisms, MIC was determined (Figure 11b). The MIC value of 0.39% (v/v) for *S. aureus* was the lowest obtained value, while the total inhibition of *L. monocytogenes* was achieved using 0.78% (v/v) of tested oil. For both tested yeasts, the MIC value is 1.56% (v/v), which suggests that eukaryotic cells are more resistant to the tested antimicrobial agent. This observation is in accordance with the obtained MIC value of 3.125% (v/v) for *A. brasiliensis*, which is eight and four times higher than for *S. aureus* and *L. monocytogenes*, respectively. When it comes to application of an antimicrobial agent into a packaging material, a very important aspect of antimicrobial activity is the time-based monitoring of the inhibition effect, that is, pharmacodynamics potential of the MIC value of LG essential oil during the contact time between the antimicrobial agent and microbial cells. Therefore, the pharmacodynamic potential was determined by the time-kill kinetics study method, following the antimicrobial effect for 24 h for sensitive bacteria (Figure 11c) and 72 h for yeasts and fungi (Figure 11d). Parallel to the set experiment, non-treated bacterial suspensions were incubated under the same incubation conditions to perceive microbial behavior without the addition of MIC values of the LG essential oil. Both graphs (Figures 11c,d) represent growth profile curves, which indicates the number of viable cells over the defined contact or incubation time. For non-treated suspensions, the obtained curves have an expected S-shape, with noticeable growth stages during microbial reproduction: lag, log, and stationary, with the increase of the cell number in the range of 1.5 to 3 log CFU/mL. On the contrary, the obtained curves for MIC values indicate a biocide effect for *S. aureus* and *L. monocytogenes* after a contact time of 8 and 12 h, respectively, while the same effect was observed for both yeasts for 18 h. A fungicide effect was achieved in a 6 h shorter time

for *A. brasiliensis*. Interestingly, the effect of MIC concentration lasted until the end of the contact time, indicating that the applied concentrations certainly have a biocidal effect on all cells. In other words, the applied concentration of LG essential oil is present during the whole contact time, which prevents injured cells to resume. In summary, it can be concluded that a contact time of 18 h between LG essential oil with a concentration of above 3.125% (v/v) and sensitive microorganisms is certainly enough for complete inhibition of all tested microorganisms present in a concentration of approx. 5 log CFU (initial concentration in the experiment).

For a deep investigation of the antimicrobial effect of the prepared CA/PCL films, antimicrobial effects during direct contact between films and cells are evaluated (Figure 12). As control films without LG essential oil did not inhibit the tested microorganisms, the obtained values of antimicrobial activity can be attributed to the LG addition and successive release during the contact time. Briefly, CA/PCL 10%, 3% LG has shown a high antimicrobial effect against yeasts, but an average or low effect against bacteria and fungi. The obtained differences can be the consequence of an insufficient amount of LG essential oil in the film, which did not release the MIC value for the biocide effect against prokaryotic cells and fungi. A similar pathway was observed for the CA/PCL 10%, 5% LG film, but with higher average values of the inhibition zone. CA/PCL 10%, 7% LG films demonstrated a high antimicrobial effect of the tested material against all tested microorganisms. For a better understanding of different levels of antimicrobial effect, Figure 10 additionally shows a difference in the inhibition zones obtained in the case of *A. brasiliensis*, where the addition of 3, 5, and 7 wt % LG shows a low, average, and high level of antimicrobial effects during the contact time, respectively. It can be concluded that using 7% LG addition in

Table 7. Assessment of the Efficiency of Fresh Fruit Packaging

Samples	CA/PCL-diol 10% control						CA/PCL-diol 10% 7% LG					
	0	1	3	5	7	10	0	1	3	5	7	10
storage period												
evaluation of the surface affected with fungal growth (%)	0	0	5	40	70	85	0	0	0	0	5	10

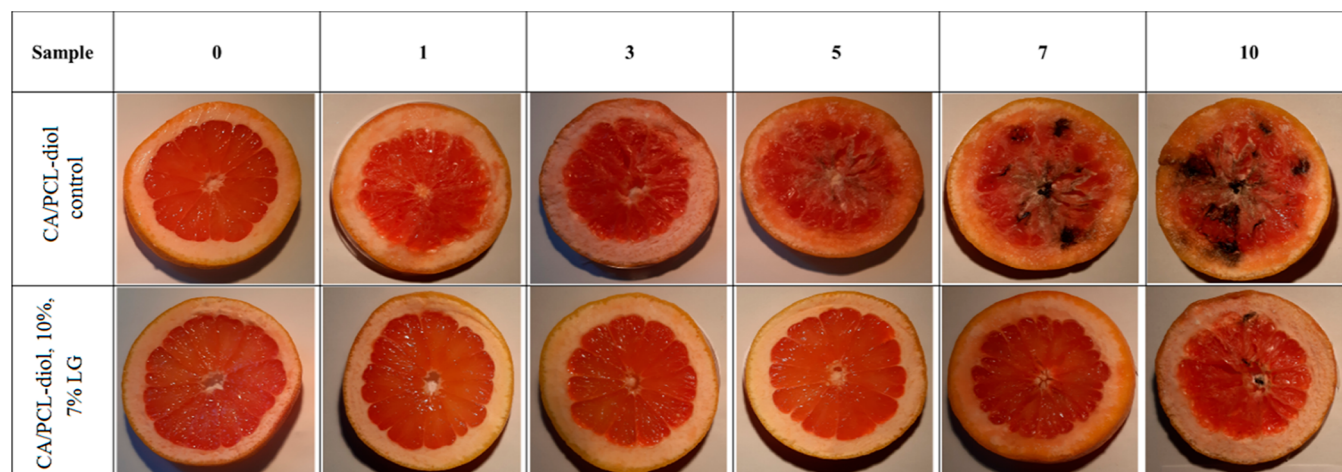


Figure 13. Photographs of grapefruit slice packed in the CA/PCL-diol control or LG film during the storage period.

films can inhibit all three types of microorganisms, and the releasing rate of antimicrobial agents is at a satisfactory level.

Preliminary Assessment of the Efficiency of Fresh Fruit Packaging. As with all other citrus, grapefruits have to be harvested in full maturity and cannot be stored at very low temperatures because the development of skin defects is almost inevitable.⁴² Additionally, one of the usual microbial contaminations of grapefruits is directed to *Aspergillus* strains, which cause a variation in the skin color and typical black mycelia on pulp and seeds.⁴³ Therefore, artificial contamination by *A. brasiliensis*, which was previously defined as sensitive to lemongrass essential oil (Figures 11 and 12), was done using freshly sliced grapefruits. According to the obtained results of biological parameters (Table 7, Figure 13), the storage period can be prolonged for approximately double time. The sample with 7 wt % of LG was selected for preliminary assessment, considering the results of the disk-diffusion test and its optimal antifungal activity. The addition of 7% lemongrass essential oil into the CA/PCL film enables resistance against fungal growth better than that of control film alone, which suggests that the used antimicrobial agent in a concentration-dependent manner is effective and remarkably reduced targeted fungi. The initial phase of fungal growth on control samples can be seen with the naked eye on the third day of storage (5% of the surface affected with visible contamination), while the same level of growth was observed on the seventh day on fruit samples packaged with the CA/PCL film with 7% LG. This preliminary examination stopped after 10 days due to the continued untenability of the fruit itself.

Lemongrass essential oil was effective and remarkably reduced fungi for 10 days of storage of grapefruits. Using LG as an antimicrobial agent incorporated into CA/PCL diol blends, control of microbial growth was achieved. The quality of grapefruits during storage has been limited by its perishable nature including specific conditions for storage and susceptibility to postharvest diseases associated with fungal contamination. In the current study, CA/PCL diol blends with 7%

LG essential oil effectively inhibited or controlled targeted fungal populations.

CONCLUSIONS

In the present work, a novel bio-based plasticizer glycerol tritartarate has been employed for the preparation of CA/PCL-diol films intended for fruit packaging. The used plasticizer has shown a compatibilizing effect, improving the miscibility of CA and PCL-diol. By addition of lemongrass essential oil in a film with optimized formulation (10 wt % of plasticizer), a framework has been made for obtaining biodegradable active packaging, considering the results of disc-diffusion method and proved the antimicrobial activity of lemongrass oil. Therefore, this sample has been used for preliminary assessment conducted with artificial contamination of grapefruit by *A. brasiliensis* because *Aspergillus* strains cause a variation in the skin color and typical black mycelia on grapefruit pulps and seeds. Wrapping the grapefruit by the film that contains 7 wt % of lemongrass oil results in a significantly greater shelf life (extended by 2.33 times), considering the fact that the film without LG oil enables preservation for 3 days, when the first changes have occurred, while the same level of fungal growth on the fruit samples packaged with the CA/PCL film with 7% LG was observed on the seventh day. The obtained results confirm the potential use of LG oil addition in the CA/PCL film and thereby provide a platform for manufacturing biodegradable and cost-effective active packaging.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acssuschemeng.2c02009>.

DSC curve of PCL-diol; DSC curve of CA; DTG curve of a neat CA; DTG curve of a neat PCL-diol; DTG of blend CA/PCL-diol 10%; and DTG of blend CA/PCL-diol 10%, 7% LG (PDF)

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Author Contributions

The manuscript was written through the contributions of all authors. All authors have approved the final version of the manuscript.

Notes

The authors declare no competing financial interest.

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